

Investigating the photocatalytic activity in bactericidal effect of iron(III) oxide bi-phase composite against *Staphylococcus aureus*

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Abstract. This work describes the phase transition process used to create a bi-phase composite (called “ $\alpha/\gamma\text{-Fe}_2\text{O}_3$ ”) from two iron precursors (FeSO_4 and FeCl_3). It was discovered that controlling the temperature between 200 and 800 °C was effective in both producing $\gamma\text{-Fe}_2\text{O}_3$ nanoparticles and initiating the phase transition of $\gamma\text{-Fe}_2\text{O}_3$ into $\alpha\text{-Fe}_2\text{O}_3$ nanoparticles. The formation of a bi-phase mixture ($\alpha/\gamma\text{-Fe}_2\text{O}_3$) occurred at 500 °C when the phase transition process was taking place. Consequently, the bactericidal efficiency of this bi-phase $\alpha/\gamma\text{-Fe}_2\text{O}_3$ composite was more excellent than that of the single-phase $\alpha\text{-Fe}_2\text{O}_3$ or $\gamma\text{-Fe}_2\text{O}_3$ material (from 1.1 to 1.6 times, respectively). Here, the bi-phase contacts probably formed between the α - and γ -phase nanoparticles of Fe_2O_3 played an important role in reducing the electron-hole recombination, and thus, increasing the photocatalytic and bactericidal efficiency of the developed composite material. Overall, our findings highlight the potential of this bi-phase material for practical photocatalytic applications.

Keywords: iron oxide, bi-phase, photocatalytic, bactericidal activity, *Staphylococcus aureus*.

Classification numbers: 2.4.2.

1. INTRODUCTION

Staphylococcus aureus (*S. aureus*) is an anaerobic bacterium frequently present on the skin, respiratory system, and nasal passages [1]. *S. aureus* enters the body by binding to surfaces like the skin and nasal cavity, aided by secretory agents [2]. This process can lead to various infectious conditions, encompassing minor skin and soft tissue infections, infective endocarditis, osteomyelitis, bacteremia, and life-threatening pneumonia [3]. It is estimated that *S. aureus* bacteremia has an incidence ranging from 20 to 50 cases per 100,000 cases of skin infections

yearly, and 10 % to 30 % of these patients die from the infection, placing a significant burden on the public health system [4]. Therefore, pre-emptive eradication of *S. aureus* prior to host attachment assumes critical significance. Currently, several methods, such as high-intensity UV-C light [5-7], chemical disinfection of surfaces [8], and antibiotics [9], are employed for *S. aureus* bactericidal. Besides, there has been a growing interest in photocatalytic activity (assisting in using antibiotics) as an alternative avenue of exploration. Scientific researchers have demonstrated the effective antibacterial properties of conventional semiconductor oxides (e.g., TiO₂ and ZnO) through photocatalysis [10, 11]. However, their wide bandgaps limit their photocatalytic functions under visible light irradiation, despite visible light constituting 43 % of the solar spectrum [12]. As a result, semiconductor oxides having small bandgaps and being active under visible light have drawn much attention, such as CuO, Bi₂O₃, WO₃, Fe₂O₃, and CdS [13]. Among them, the Fe₂O₃ is of great interest due to its abundance, inexpensiveness, ease of synthesis, non-toxicity, and chemical stability [14]. Especially, the Fe₂O₃ can appear in four different phases: α -Fe₂O₃, β -Fe₂O₃, γ -Fe₂O₃, and ϵ -Fe₂O₃ [15]. Notably, the bandgap values of α -Fe₂O₃ and γ -Fe₂O₃ range from 2.0 to 2.4 eV, which are appropriate for maximizing the photocatalytic activity under visible light [16, 17].

Recently, Pallela *et al.* [18] and Rana *et al.* [19] have reported the antibacterial efficacies of α -Fe₂O₃ and γ -Fe₂O₃ against *S. aureus*, respectively. Their proposed mechanism for bacterial eradication involves the generation of reactive oxygen species (ROS), including hydroxyl radicals (\cdot OH), hydrogen peroxide (H₂O₂), and superoxide (\cdot O²⁻), when α -Fe₂O₃ and γ -Fe₂O₃ were excited under visible light. That generated free radicals, which can disrupt polysaccharide activity, inactivate enzymes, induce DNA fragmentation, and initiate lipid peroxidation processes, ultimately leading to bacterial death. The fast recombination of electrons and holes in the single-phase γ -Fe₂O₃ or α -Fe₂O₃ material is an issue for its photocatalytic activity. It was followed by a shortage of electron and hole migrations to the material's surface, resulting in a continuous decline in the ROS production and thereby decreasing the antibacterial efficacy [20]. To address this problem, researchers have proposed employing heterojunctions to eliminate electron and hole recombination and enhance photocatalytic efficacy [21, 22]. However, to form either junctions or contacts based on the γ -Fe₂O₃ or α -Fe₂O₃, it is often necessary to utilize different types of materials, which can potentially result in alterations to the properties of Fe₂O₃. Interestingly, the Fe₂O₃ possesses a phase transition from γ -Fe₂O₃ to α -Fe₂O₃ at high temperatures. By controlling the phase transition temperature, a bi-phase contact between the α - and γ - phases can be formed, and it is promising for reducing electron-hole recombination and enhancing photocatalytic efficiency [23, 24].

In this work, we developed the bi-phase composite based on α - and γ -Fe₂O₃, and we investigated not only its photocatalytic activity but also its antibacterial activity against *S. aureus*. Through characterizations, the bactericidal ability of single-phase α -Fe₂O₃ or γ -Fe₂O₃ samples was found to be 51.54 % or 76.67 %, respectively. Meanwhile, the bi-phase α/γ -Fe₂O₃ samples exhibited outstanding bactericidal efficiency of up to 84.62 % under the same testing conditions of material utilization. This finding confirms the potential and effectiveness of replacing conventional single-phase materials with bi-phase materials in order to prevent electron-hole recombination for photocatalysis applications.

2. EXPERIMENTAL METHOD

2.1. Fabrication of α/γ -Fe₂O₃ bi-phase mixture

The synthesis procedure is illustrated in Figure 1. Briefly, 40 mL of a 0.25 M FeSO₄ solution was prepared by adding FeSO₄·7H₂O (Sigma-Aldrich, purity \geq 98 %) to distilled water and followed by heating at 70 °C. The solution was then filled with 40 mL of 0.5 M FeCl₃·H₂O (Sigma-Aldrich, purity 97 %) so that the molar ratio of 2Fe³⁺:Fe²⁺ was reached. After stirring for 15 min, 30 mL of 5 M NaOH was gradually added to the solution until black precipitates were observed. The solution was further stirred for 60 min, then centrifuged, and washed several times with distilled water and ethanol to obtain Fe₃O₄ powder. In the final step, the prepared samples were calcined at various temperatures, including 200, 500, and 800 °C for 3 hours in the atmosphere, to produce the structures of γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃, respectively.

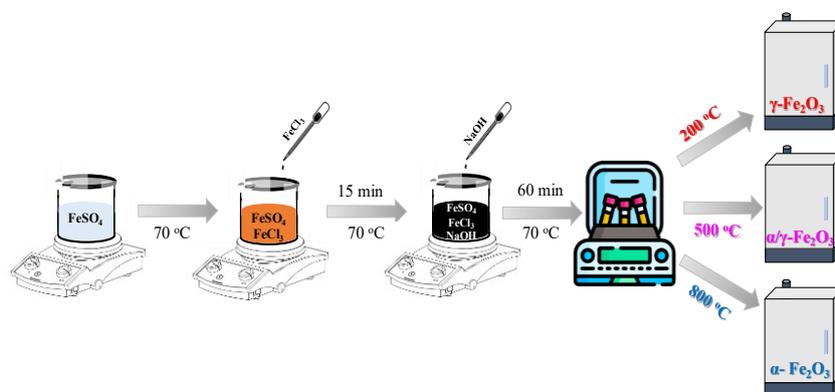


Figure 1. Synthesis process of γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃.

2.2. Characterization

The crystalline structure of synthesized samples was analyzed using X-ray diffraction (XRD-D8 Advanced) with monochromatic Cu-K α radiation ($\lambda = 1.54 \text{ \AA}$) and Raman spectroscopy (Horiba XploRA Plus) with a 532 nm laser excitation source. The material's morphology and optical properties were investigated using Scanning Electron Microscopy (SEM-JEOL JSM-7600F) and Ultraviolet-Visible spectroscopy (DS5 Edinburgh Instrument), respectively.

2.3. Bactericidal test

The bacterial strain used in this study is Gram-positive *S. aureus* ATCC 29213. The bacteria stored at -80 °C were thawed and activated by plating on TSA agar at 37 °C overnight. To ensure the purity of the strains, the *S. aureus* bacterial colony was selected, inoculated in TSB broth, and incubated at 37 °C for 24 hours. Subsequently, the bacterial suspension was diluted with phosphate-buffered saline (PBS) of 0.8 % NaCl, and its optical density was measured at a wavelength of 600 nm (OD₆₀₀). An OD₆₀₀ value of 0.07 corresponded to a bacterial density of approximately 10⁶ CFU/mL (colony-forming units per milliliter). The developed materials (γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃) were prepared at a concentration of 0.5 mg/mL, the test bacteria were added at an approximate density of 10⁶ CFU/mL, and then the mixtures were exposed to blue light (460 nm, intensity of 80 W/m²) for 180 minutes. Bacterial

density and bactericidal efficacy were determined after 30 min intervals using the serial dilution spotting method [25]. The same experiments were repeated three times. The bacterial density at each time point was determined using Eq. (1), where A_i represents the average colony count, D_i denotes the dilution factor, and V indicates the volume of each droplet in milliliters (mL).

$$N_i (\text{CFU/mL}) = A_i(D_i/V) \quad (1)$$

The bactericidal effect at each time point was calculated using Eq. (2), where N_0 represents the initial bacterial, N_i denotes the bacterial density at time point t .

$$\text{Efficiency (\%)} = (N_0 - N_i)/N_0 \times 100 \% \quad (2)$$

3. RESULTS AND DISCUSSION

3.1. Structural characteristics

The phase composition and crystal structures of the as-prepared samples $\gamma\text{-Fe}_2\text{O}_3$, $\alpha/\gamma\text{-Fe}_2\text{O}_3$, and $\alpha\text{-Fe}_2\text{O}_3$ were investigated by XRD pattern, as shown in Figure 2(a). The XRD pattern of $\gamma\text{-Fe}_2\text{O}_3$ demonstrated a cubic crystal structure (JCPDS# 39-1356) with five characterized peaks at 30.5, 35.7, 43.4, 57.6, and 63.3°, corresponding to the lattice planes (2 2 0), (3 1 1), (4 0 0), (5 1 1), and (4 4 0), respectively [26]. Meanwhile, the XRD peaks of $\alpha\text{-Fe}_2\text{O}_3$ appear at 2θ values of 24.1, 33.2, 35.8, 40.8, 49.5, 54.2, 57.7, 62.6, and 64.1°, corresponding to the lattice planes of (012), (104), (110), (113), (024), (116), (018), (214), and (300) of the rhombohedral crystal structure (JCPDS# 33-0664) [15, 27]. Interestingly, the X-ray diffraction analysis of the $\alpha/\gamma\text{-Fe}_2\text{O}_3$ sample distinctly reveals the co-occurrence of diffraction peaks, providing compelling evidence for the presence of both the $\gamma\text{-Fe}_2\text{O}_3$ and $\alpha\text{-Fe}_2\text{O}_3$ phases. Notably, the peak at about 35.7° in the XRD pattern of $\alpha/\gamma\text{-Fe}_2\text{O}_3$ is observed with a significantly broad base but a sharp tip. In the literature, this peak is known as an overlap of the (311) lattice plane of $\gamma\text{-Fe}_2\text{O}_3$ and the (110) lattice plane of $\alpha\text{-Fe}_2\text{O}_3$ [23]. Therefore, to precisely ascertain the composition positions of the diffraction peaks corresponding to the (311) lattice plane of $\gamma\text{-Fe}_2\text{O}_3$ and the (110) lattice plane of $\alpha\text{-Fe}_2\text{O}_3$, we performed peak deconvolution analysis on the XRD pattern of the $\alpha/\gamma\text{-Fe}_2\text{O}_3$ sample at approximately 35.7°. The result of this analysis is depicted in Figure 2(b), and this revealed the difference from the peak positions of the (311) lattice plane of $\gamma\text{-Fe}_2\text{O}_3$ (at 35.69°) and the (110) lattice plane of $\alpha\text{-Fe}_2\text{O}_3$ (at 35.79°) in the $\alpha/\gamma\text{-Fe}_2\text{O}_3$ composite. Our deconvolution approach and findings here are in complete agreement with another study on the peak positions of $\alpha\text{-Fe}_2\text{O}_3$ and $\gamma\text{-Fe}_2\text{O}_3$ within the $\alpha/\gamma\text{-Fe}_2\text{O}_3$ bi-phase sample reported by Bi, Tao, Ren, Huang, WangHao [23]. Overall, these analyses confirm that there is a bi-phase mixture in our $\alpha/\gamma\text{-Fe}_2\text{O}_3$ sample.

The calculation for determining the proportion of $\alpha\text{-Fe}_2\text{O}_3$ and $\gamma\text{-Fe}_2\text{O}_3$ is performed using Eq. (3), where I_γ and I_α represent the diffraction intensities of the (311) plane of $\gamma\text{-Fe}_2\text{O}_3$ and the (104) plane of $\alpha\text{-Fe}_2\text{O}_3$ in the $\alpha/\gamma\text{-Fe}_2\text{O}_3$ sample, we were able to determine the phase percentages of $\gamma\text{-Fe}_2\text{O}_3$ and $\alpha\text{-Fe}_2\text{O}_3$ to be 44 and 56 %, respectively [23, 24].

$$\% \alpha\text{-Fe}_2\text{O}_3 = 1/(1+1.7I_\gamma/I_\alpha) \quad (3)$$

In addition, to calculate the average crystallite size Scherrer's formula was used, as indicated by Eq. (4) [12], where D is the average crystallite size, λ is the wavelength (1.541 Å), θ is the Bragg diffraction angle, and β is the full-width half maximum (FWHM).

$$D = 0.9\lambda/[\beta\cos(\theta)] \quad (4)$$

Crystalline parameters were calculated from XRD data and Scherrer's formula by considering the prominent peak at (311) for γ -Fe₂O₃ and (110) for α -Fe₂O₃ in Table 1. The results showed that the γ -Fe₂O₃ sample had an initial crystallite size of about 9 nm. In the α/γ -Fe₂O₃ sample, the crystallite size of γ -Fe₂O₃ increased to 14.71 nm and α -Fe₂O₃ was about 30.26 nm. At 800 °C, the γ phase completely transformed into α -Fe₂O₃ and the α -Fe₂O₃ crystallite size increased to about 34.96 nm. It can be seen that the crystallite size increases as the calcination temperature increases.

Table 1. Crystalline parameters were calculated using XRD data and Scherrer's formula.

Sample	Phase	Crystal lattice	2 θ position (°)	FWHM β (rad)	Crystallite size (nm)
γ -Fe ₂ O ₃	γ	(3 1 1)	35.7	0.927	9.000
α/γ -Fe ₂ O ₃	γ	(3 1 1)	35.69	0.569	14.709
	α	(1 0 4)	35.79	0.275	30.266
α -Fe ₂ O ₃	α	(1 0 4)	35.8	0.238	34.962

The Raman analysis in Figure 2(c) reveals the characteristic peaks of γ -Fe₂O₃ at positions of 380, 490, and 680 cm⁻¹ [28, 29]. At the same time, the α -Fe₂O₃ sample exhibits the characteristic peaks at 222, 291, and 406 cm⁻¹ [30]. Particularly, the α/γ -Fe₂O₃ sample demonstrates discernible oscillatory behavior at 216, 280, and 390 cm⁻¹. This analysis reveals a pronounced blue shift in the oscillation of the α/γ -Fe₂O₃ sample compared to α -Fe₂O₃, which is similar to a recent publication by Tokubuchi *et al.* [31].

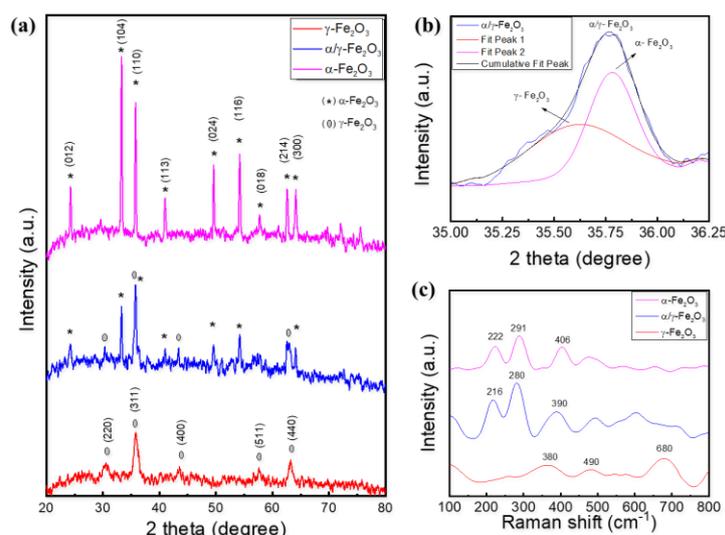


Figure 2. (a) XRD patterns of γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃; (b) Deconvolution of α/γ -Fe₂O₃ characteristic peak at $\sim 35.75^\circ$ to find its composition peaks, which are γ -Fe₂O₃ (Fit peak 1) and α -Fe₂O₃ (Fit peak 2); (c) Raman spectra of those samples.

3.2. Morphology analysis

The SEM image of the γ -Fe₂O₃ sample in Figure 3(a) shows the presence of γ -Fe₂O₃ nanoparticles ranging in size from 20-30 nm. In Figure 3(c), the α -Fe₂O₃ sample exhibited a coral-like morphology, similar to previous reports on α -Fe₂O₃ synthesis using the phase transition method [32]. The length and width of the α -Fe₂O₃ material are approximately 150-200

nm and 50-100 nm, respectively. The structural change in the crystal lattice of γ -Fe₂O₃ samples was seen at high temperatures [33]. Specifically, the cubic crystal structure of γ -Fe₂O₃ transforms into the rhombohedral one of α -Fe₂O₃, resulting in significant alterations in the morphological characteristics [34]. Reasonably, Figure 3(b) presents an SEM image of the calcined α/γ -Fe₂O₃ sample at 500 °C. At that temperature, the γ -Fe₂O₃ structure has incompletely converted to α -Fe₂O₃. Consequently, a coexistence of γ -Fe₂O₃ particles adhering to the network nodes of α -Fe₂O₃ is observed; here, Figures 3(d)-(f) reveal that there are contacts between the two types of Fe₂O₃ nanoparticles, which can be considered as bi-phase contacts [23, 35].

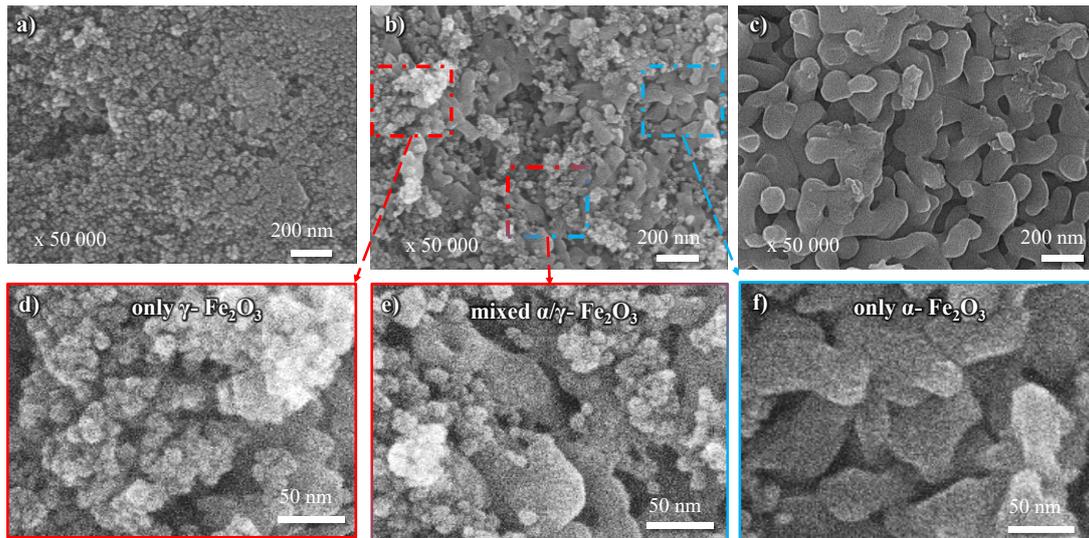


Figure 3. (a-c) SEM images of iron oxide materials produced at different temperatures: (a) γ -Fe₂O₃ after 200 °C; (b) α/γ -Fe₂O₃ after 500 °C; (c) α -Fe₂O₃ after 800 °C; (d)-(f) Magnified areas from the SEM image of α/γ -Fe₂O₃ after 500 °C, which show (left) a cluster of only γ -Fe₂O₃ nanoparticles, (middle) a cluster of α - and γ -Fe₂O₃ nanoparticles (i.e., bi-phase contact), and (right) a cluster of only α -Fe₂O₃.

3.3. Optical properties

The optical characteristics of the γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃ samples are represented by the UV-Vis spectra in Figure 4(a). The spectral analysis reveals absorption edges at around 575 nm and 740 nm for the γ -Fe₂O₃ and α -Fe₂O₃ samples, respectively. The absorption curve shapes of both γ -Fe₂O₃ and α -Fe₂O₃ found in this experiment closely resemble the shape reported in prior publications [36, 37]. Interestingly, the combination in the bi-phase α/γ -Fe₂O₃ material has led to changes in the shape of the absorption edge and peak compared to the pure γ -Fe₂O₃ or pure α -Fe₂O₃. In the literature, the optical absorption edge of an active material can be shifted due to the formation of a new phase or the phase transition of that material [23, 38-40]. According to our observation, the shifted absorption edge in our α/γ -Fe₂O₃ sample can be attributed to the following reasons, including the α phase formation in the γ -Fe₂O₃ nanoparticle matrix, as well as the phase transition of iron oxide from the γ phase to the α phase. In general, due to the structural changes (incompletely), the optical absorption spectrum of our bi-phase composite is slightly different from that of mono-phase Fe₂O₃ materials. Here, for further confirmation, we utilized the Tauc's method to estimate the optical band gap of our materials, as expressed in Eq. (5); where α , h , ν , B , and E_g are the absorption coefficient, Planck constant, light frequency, proportionality constant, and bandgap, respectively. Figure 4(b) presents the

$(ah\nu)^2$ versus $h\nu$ plot, showing the bandgap energy of γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃ to be 2.32, 2.24, and 2.02 eV, respectively. This result implies that α/γ -Fe₂O₃ has a high photocatalytic potential in the visible light range [32].

$$(ah\nu)^2 = B(h\nu - E_g) \quad (5)$$

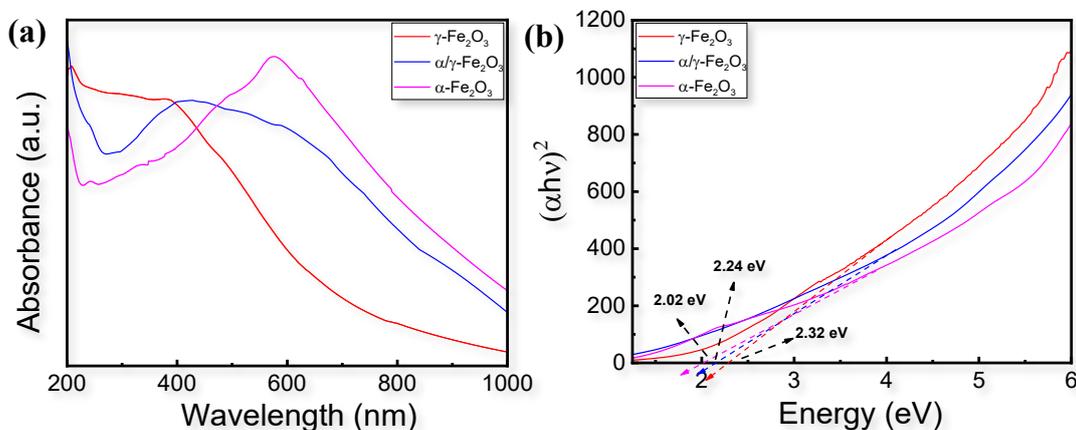


Figure 4. (a) UV-Vis absorption spectra and (b) Tauc plots $[(ah\nu)^2 vs h\nu]$ of γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃ materials.

3.4. Antibacterial activity

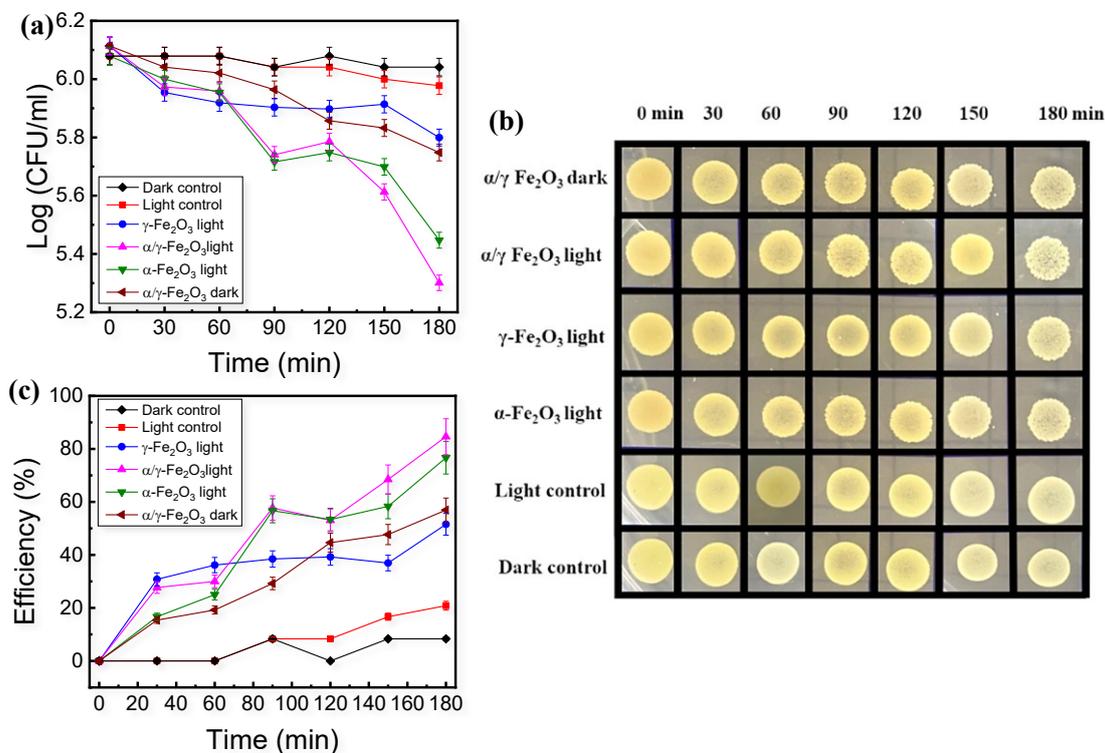


Figure 5. (a) Evaluation of the bacterial survivalability (Log (CFU/mL)) of three samples under light and dark conditions; (b) Photoimages showing their colonies in 10 μ L over time; (c) Bactericidal efficiency over time against *S. aureus* bacterium when exposed to γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃.

Figure 5(a) shows the bacterial survival ability of the *S. aureus* ($\sim 10^6$ CFU/mL) when exposed to γ -Fe₂O₃, α/γ -Fe₂O₃, or α -Fe₂O₃ at a concentration of 0.5 mg/mL for 180 minutes. After 30 min intervals, the samples were sequentially diluted with PBS (in 0.8 % NaCl solution) to achieve 10^{-1} , 10^{-2} , or 10^{-3} decimal concentrations, and so on, depending on the bacterial sample. Each dilution step by 10^{-1} was carried out by adding 1 mL of the sample (or solution from the previous dilution) into 9 mL of PBS in a test tube. The diluted samples were then dropped onto TSA agar plates and incubated at 37 °C for 24 hours. From the result of colony counting, the respective number is averaged. Figure 5(b) presents the counts of colonies in 10 μ L over time. The α/γ -Fe₂O₃ material demonstrates a superior photocatalytic efficiency under illumination, resulting in the lowest bacterial density observed. Figure 5(c) shows the bactericidal effect of γ -Fe₂O₃, α/γ -Fe₂O₃, and α -Fe₂O₃ materials over time against *S. aureus*, which is confirmed by Eq. (2). These results indicate that α/γ -Fe₂O₃ exhibits a superior antibacterial ability compared to α -Fe₂O₃ and γ -Fe₂O₃. After 180 minutes of illumination, α/γ -Fe₂O₃ reduced the bacterial density of *S. aureus* from $6.11 \log_{10}$ to $5.30 \log_{10}$, achieving a bactericidal efficiency of 84.62 %. Meanwhile, the disinfectant abilities of γ -Fe₂O₃ and α -Fe₂O₃ were 51.54 and 76.67 %, respectively. The bactericidal efficiency of the bi-phase α/γ -Fe₂O₃ is 1.1 times higher than that of α -Fe₂O₃ and upto 1.6 times higher than that of the γ -Fe₂O₃.

3.5. Proposed bactericidal mechanism of α/γ -Fe₂O₃ composite

To explain the improved photocatalytic and bactericidal ability of our α/γ -Fe₂O₃ composite, we based on the electronic band structure to propose its mechanism. Particularly, the band edge positions of the conduction band (CB) and valence band (VB) of α -Fe₂O₃ and γ -Fe₂O₃ were estimated using empirical equations [41], noted as Eqs. (6) and (7).

$$E_{CB} = X - E_e - 0.5E_g \quad (6)$$

$$E_{VB} = E_{CB} + E_g \quad (7)$$

Here, E_e is the energy of free electrons on the hydrogen scale ($E_e = 4.5$ eV), and X is the absolute electronegativity of a semiconductor. The X values for α -Fe₂O₃ and γ -Fe₂O₃ are 4.78 and 5.87 eV, respectively [41, 42]. E_g is the band gap energy of the semiconductor, 2.02 eV for α -Fe₂O₃ and 2.32 eV for γ -Fe₂O₃. From these calculations, the E_{CB} values of α -Fe₂O₃ and γ -Fe₂O₃ were determined to be ca. -0.73 and 0.21 eV versus *NHE*, respectively, whereas the E_{VB} values were estimated to be 1.29 and 2.53 eV versus *NHE*. The bactericidal mechanism of the α/γ -Fe₂O₃ mixture is described in Figure 6. Under blue-light irradiation ($\lambda = 460$ nm), photo-induced free electrons and holes were generated inside the α/γ -Fe₂O₃ composite. At bi-phase contacts (between α - and γ - phases), electrons from the conduction band (CB) of α -Fe₂O₃ migrate to the CB of γ -Fe₂O₃ due to the potential difference. Meanwhile, holes move from the valence band (VB) of γ -Fe₂O₃ to the VB of α -Fe₂O₃. As a result, the electron-hole recombination mechanism in the bi-phase structure is significantly weakened [23, 24]. It is important to note that the CB energy level of γ -Fe₂O₃ (approximately 0.21 eV) is not sufficient to reduce O₂ into $\cdot\text{O}_2^-$, as the reduction potential of O₂/ $\cdot\text{O}_2^-$ is -0.33 eV. Consequently, O₂ would directly react with photo-induced free electrons to produce H₂O₂, as illustrated in Eq. (8) [43]. On the other hand, the holes (h^+) in the VB of α -Fe₂O₃ may contribute to bacterial inactivation through direct oxidation of the cell membrane because their energy (approximately 1.29 eV) are

not sufficiently oxidizing to produce $\cdot\text{OH}$ radicals [43]. The oxidation potentials for $\text{OH}^-/\cdot\text{OH}$ and $\text{H}_2\text{O}/\cdot\text{OH}$ are 1.99 eV and 2.34 eV vs NHE, respectively [44].



In addition, Fenton reactions can occur due to the reductive dissolution of the material surface in aqueous environments, leading to the production of Fe^{2+} ions. These ions can then be oxidized in the presence of H_2O_2 to form Fe^{3+} , causing the decomposition of H_2O_2 into hydroxyl radicals [14], as illustrated in Eq. (9). Therefore, reactive oxygen species (ROS), predominantly comprising hydroxyl radicals ($\cdot\text{OH}$) and hydrogen peroxide (H_2O_2) interact with various components within bacterial cells, resulting in the destruction and demise of bacterial cells [18].



Besides, taking advantage of the thermally induced structural changes, particularly the formation of the bi-phase $\alpha/\gamma\text{-Fe}_2\text{O}_3$ contact, our developed composite demonstrates more effective charge carrier separation and consequently ROS generation efficiency than single-phase materials.

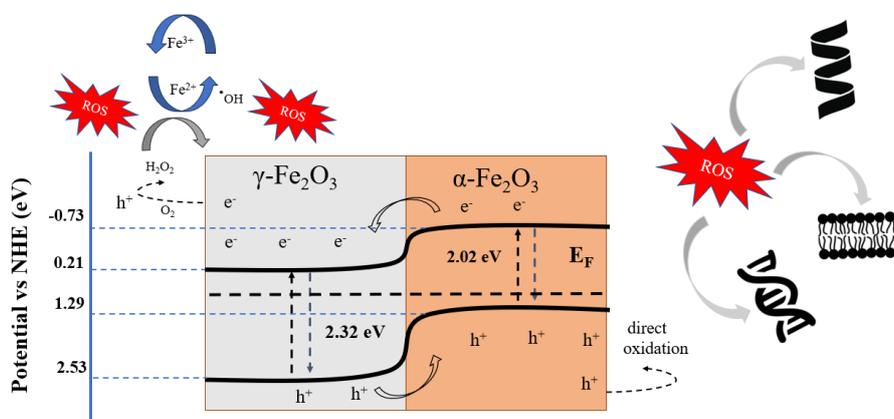


Figure 6. Bactericidal mechanism of $\alpha/\gamma\text{-Fe}_2\text{O}_3$ mixture.

4. CONCLUSIONS

We have successfully developed a straightforward phase-transition technique for the synthesis of $\alpha/\gamma\text{-Fe}_2\text{O}_3$ (a bi-phase composite). Due to the temperature-induced structural changes, especially the formation of bi-phase contacts between γ - and α - phases, this material shows the significantly reduced electron-hole recombination and improved photocatalytic efficiency. According to the findings, $\alpha/\gamma\text{-Fe}_2\text{O}_3$ can kill bacteria with an excellent efficiency of up to 84.62 % within 180 minutes. As compared to the single-phase materials, the $\alpha/\gamma\text{-Fe}_2\text{O}_3$ bi-phase mixture has a bactericidal efficiency about 1.1 times and 1.6 times higher than that of $\alpha\text{-Fe}_2\text{O}_3$ and $\gamma\text{-Fe}_2\text{O}_3$, respectively. This critical finding implies that our bi-phase mixture has the potential to replace conventional single-phase materials in the field of photocatalysis.

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CRedit authorship contribution statement. Tan Muon Dinh: Methodology, Investigation. Nguyen Thanh Loan: Resources, Investigation. Hoang Hung Nguyen: Formal analysis, Writing – original draft. Tran Viet Cuong: Formal analysis, Writing – original draft. Vinh Quang Dang: Formal analysis, Writing – original draft. Huynh Tran My Hoa: Conceptualization, Supervision, Writing – review & editing.

Declaration of competing interest. The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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