

Synthesis and properties of g-C₃N₄ bulk and nanosheet from thiourea

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Received: 5 September 2023; Accepted for publication: 5 June 2025

Abstract. Graphitic carbon nitride (g-C₃N₄) is a metal-free polymer semiconductor of interest for visible-light photocatalysis. Here, the g-C₃N₄ bulk was prepared by thermal polycondensation of thiourea in air (480–550 °C, 2 h; 5 °C min⁻¹). XRD and FTIR analysis confirmed the heptazine/triazine framework, with well-defined (002) and (100) reflections obtained at 520–550 °C, while lower temperatures showed incomplete condensation. Few-layer g-C₃N₄ nanosheets were then produced by thermal oxidation/exfoliation of the bulk at 500 °C in static air, yielding an expanded, porous morphology observed by FESEM and a weaker (002) peak together with an additional feature near 23°. Optical characterization (UV–Vis DRS and PL) indicated improved light utilization and charge-carrier behavior for the nanosheets: the optical bandgap decreased from 2.750 eV (bulk) to 2.625 eV (nanosheets) and the PL maximum blue-shifted (≈465 → 444 nm) with lower intensity, suggesting suppressed electron–hole recombination. Lorentz deconvolution of PL spectra revealed three main emission centers associated with distinct transition pathways. This simple, solvent-free route using low-cost thiourea provides scalable g-C₃N₄ bulk and nanosheets for visible-light-driven environmental and energy applications.

Keywords: graphitic carbon nitride, g-C₃N₄ nanosheet, thiourea precursor, polymerization, photocatalytic materials.

Classification numbers: 2.4.2, 2.1.3.

1. INTRODUCTION

Graphitic carbon nitride (g-C₃N₄) is the most stable allotrope of carbon nitride (C₃N₄) at room temperature [1]. Owing to its high thermal and chemical stability, low-cost and abundant elements, environmental friendliness, and suitable band structure aligned with water redox potentials, g-C₃N₄ has been widely studied for photocatalytic applications [2, 3]. It exhibits activity in organic pollutant degradation [4], water splitting for H₂ and O₂ production [5], and CO₂ reduction to hydrocarbon fuels [6]. However, its photocatalytic performance is limited by

the high recombination rate of photogenerated electron–hole pairs and a low density of active sites [7].

Two-dimensional (2D) *g*-C₃N₄ nanosheets exhibit a large specific surface area, which enhances reactive sites, accelerates mass transfer, and improves light absorption. Compared with bulk *g*-C₃N₄, 2D nanosheets also promote charge separation and prolong carrier lifetime [8–14]. Various synthesis strategies have been reported, including in situ polymerization methods [15], thermal exfoliation [16,17], chemical exfoliation [18], and plasma-assisted techniques [19].

In this work, the bulk *g*-C₃N₄ was synthesized via single-step thermal polycondensation of thiourea. The influence of thiourea precursors on structural formation and properties was investigated. Furthermore, nanosheets were obtained by thermal exfoliation of the bulk material in air to evaluate the structural and optical modifications induced by exfoliation.

2. MATERIALS AND METHODS

2.1. Materials

Thiourea was purchased from Sigma-Aldrich with analytical grade purity and used without further purification.

2.2. Synthesis of *g*-C₃N₄ bulk and nanosheets

The *g*-C₃N₄ bulk was synthesized by a thermal polymerization method using thiourea precursors. In a typical synthesis, 10 g of thiourea were placed in a porcelain crucible with a cover and heated at temperatures of 480, 500, 520, and 550 °C in air at a rate of 5 °C/min for 2 hours. After cooling to room temperature, the resulting pale-yellow product was collected.

Nanosheets were prepared by thermal exfoliation of the obtained *g*-C₃N₄ bulk. Specifically, 0.5 g of the bulk sample were placed in an open porcelain crucible and heated at 500 °C in air for 2 hours at a rate of 5 °C/min. The resulting light-yellow powder was collected as *g*-C₃N₄ nanosheets. Figure 1 presents the diagram of the synthesis of *g*-C₃N₄ bulk and nanosheet.

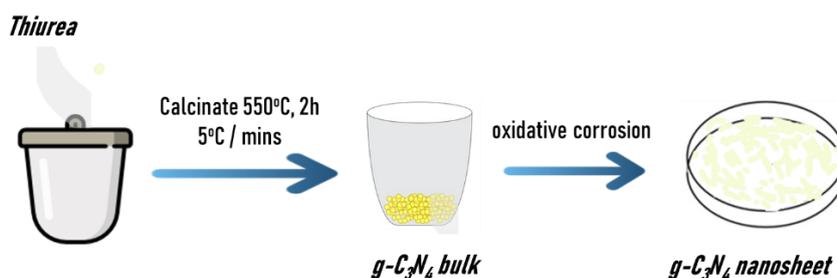


Figure 1. Diagram of synthesis of *g*-C₃N₄ bulk and nanosheet.

2.3. Characterization

The crystal structure and phase purity were obtained using X'pert Pro (PANalytical) MPD with CuK- α 1 radiation (= 1.54056 Å) at a scanning rate of 0.03°/2s in the 2 θ range from 10° to 70°. The crystal analysis was performed by HighScore Plus software using the ICDD database. FTIR spectra were analyzed using FT/IR-4600 type A (JASCO) with a wavenumber range from

500 cm^{-1} to 4000 cm^{-1} . The morphology of the samples was investigated using a HITACHI S4800 field emission scanning electron microscope (FESEM, Japan). The diffusion reflectance of the samples was measured on a JASCO V-750 instrument using a 60 mm ISV-922 integrating sphere with a scanning rate of 200 nm/min and a UV-vis bandwidth of 0.50 nm. The photoluminescence (PL) properties of the $\text{g-C}_3\text{N}_4$ samples were measured.

3. RESULTS AND DISCUSSION

3.1. Structural characterization of the material

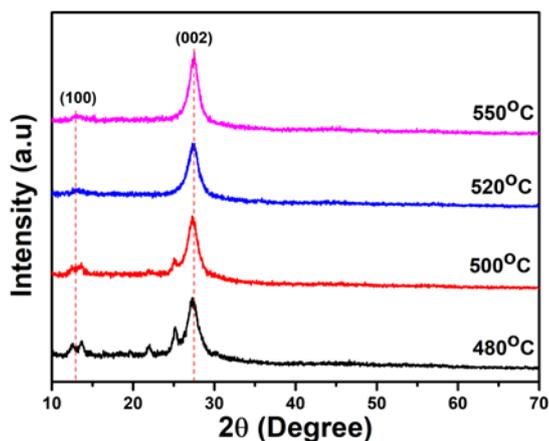


Figure 2. XRD patterns of $\text{g-C}_3\text{N}_4$ polymerization at different temperatures.

Figure 2 shows the XRD patterns of $\text{g-C}_3\text{N}_4$ synthesized by polymerization method from thiourea. The results showed that all samples synthesized exhibited the appearance of two diffraction peaks, in which a strong diffraction peak at angle $2\theta = 27.40^\circ$ corresponds to the diffraction planes (002), characterizing the superposition of aromatic conjugate systems; Another weaker diffraction peak at $2\theta = 12.78^\circ$ corresponds to the diffraction surface (100), which characterizes the cyclic arrangement of tri-s-triazine or heptazine structural units. Furthermore, the samples synthesized at 500 $^\circ\text{C}$ and 480 $^\circ\text{C}$ showed the presence of other diffraction peaks indicating incomplete formation of $\text{g-C}_3\text{N}_4$ for thiourea. The samples at 520 $^\circ\text{C}$ and 550 $^\circ\text{C}$ show that the diffraction peaks of $\text{g-C}_3\text{N}_4$ are consistent with the standard tag JCPDS number 87-1526 of $\text{g-C}_3\text{N}_4$ and previously published results [1, 20].

To investigate the vibrations of organic functional groups, the samples were analyzed using the FTIR method. FTIR spectra of $\text{g-C}_3\text{N}_4$ polymerization at different temperatures are presented in Figure 3 in the range of 500 - 4000 cm^{-1} . Typical absorption bands related to the vibration characteristic of $\text{g-C}_3\text{N}_4$ appeared in all samples. The NH stretching vibration mode from the surface uncondensed amine groups is assigned to the broadband between 3500 cm^{-1} and 3000 cm^{-1} . Bands between 1700 ~1000 cm^{-1} are assigned to the C=N extension bonds. Prolonged valence oscillations of C-N bonds in aromatic conjugate rings at 1569 cm^{-1} to 1247 cm^{-1} . The two peaks located between 806 and 884 cm^{-1} are derived from the binding of triazine units, indicating that the molecular structure of the prepared $\text{g-C}_3\text{N}_4$ consists of triazine units [21]. The results agreed well with the XRD results.

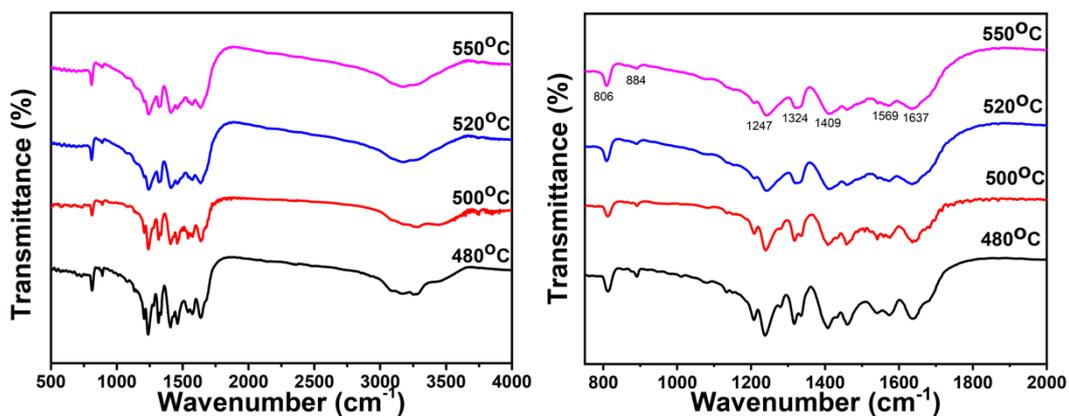


Figure 3. FTIR spectra of $g\text{-C}_3\text{N}_4$ calcination at different temperatures.

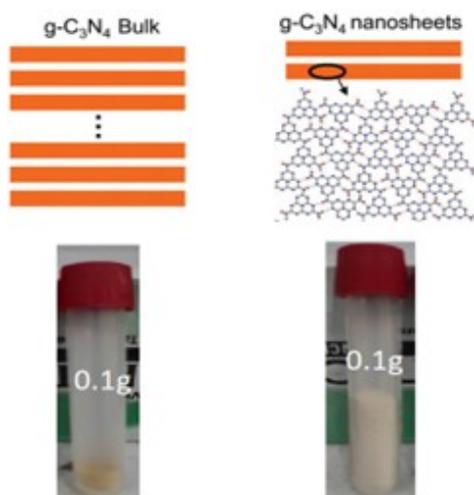


Figure 4. Images of $g\text{-C}_3\text{N}_4$ bulk and nanosheet.

The images of the $g\text{-C}_3\text{N}_4$ bulk and nanosheet are given in Figure 4. The volume of the nanosheets with the same weight is much larger than that of the bulk $g\text{-C}_3\text{N}_4$, indicating the fluffy state of the nanosheets.

3.2. Morphological characterization of the material

The morphology of $g\text{-C}_3\text{N}_4$ bulk and nanosheets were investigated with FESEM as shown in Figure 5. Compared to their parent bulk material consisting of solid agglomerates with a size of several micrometers (Figure 5(a)), the representative nanosheets appear as loose and soft agglomerates with a size of tens of micrometers (Figure 5(b)). This can be easily understood due to the gradual oxidation decomposition of the strands of polymeric melon units in the layers of bulk $g\text{-C}_3\text{N}_4$ during etching.

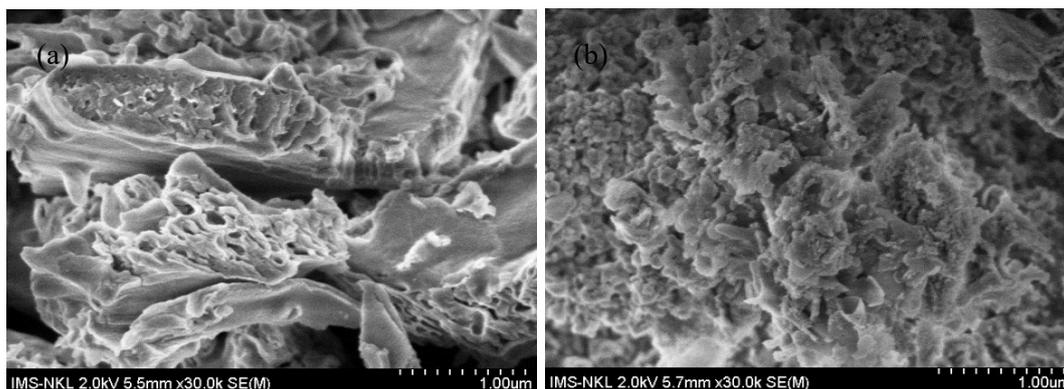


Figure 5. SEM images of g-C₃N₄: (a) bulk and (b) nanosheet.

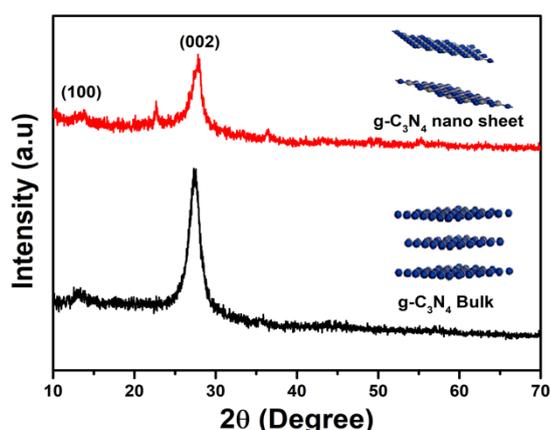


Figure 6. XRD patterns g-C₃N₄ bulk and nanosheet.

Figure 6 shows the XRD patterns of the g-C₃N₄ nanosheet and bulk samples. The results showed that the nanosheet samples exhibited weaker diffraction peak intensity than bulk samples. All samples synthesized from these two precursors showed the appearance of two diffraction peaks, in which a strong diffraction peak at the angle $2\theta = 27.4^\circ$ corresponds to the diffraction surfaces (002), characterizing the superposition of aromatic conjugate systems; Another weaker diffraction peak at $2\theta = 12.78^\circ$ corresponds to the diffraction surface (100), which characterizes the cyclic arrangement of tri-s-triazine or heptazine structural units. In addition, the nanosheet sample has an additional peak at 23° that can be explained by the presence of O₂ when performing the oxidative 'corrosion' process.

3.3. Optical properties of the material

Figure 7 presents the diffuse reflectance spectrum of the g-C₃N₄ bulk and nanosheet samples. In Figure 7(a), it is easy to see that there is a shift in the absorption amplitude towards the short wavelength. Absorption peaks at about 300 - 400 nm characterize the $\pi - \pi^*$ transition in conjugated ring systems, including heterocyclic aromatics. The features near 500 nm are due to the $n - \pi^*$ transition involving free pairs on the N atoms of the triazine ring.

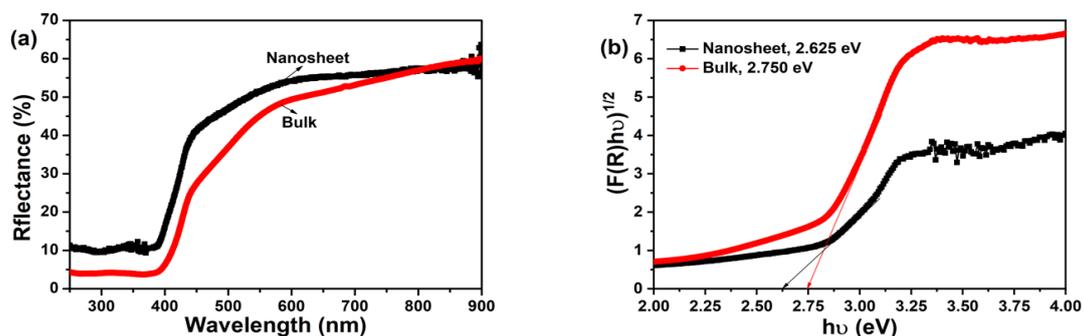


Figure 7. (a) Diffuse reflectance spectra and (b) plot $(F(R)hv)^{1/2}$ vs $(h\nu)$ of $g\text{-C}_3\text{N}_4$ bulk and nanosheet.

The optical band gap is extrapolated through the Kubelka-Munk method, where the relationship between incident photon energy $h\nu$ and the Kubelka-Munk function $F(R)$ follows:

$$(F(R).h\nu = B^*(h\nu - E_g)^n \quad (1)$$

where $F(R)$ is a Kubelka-Munk function determined from the diffuse reflectance R through the formula $F(R) = (1-R)^2/2R$; $h\nu$ is incident photon energy; and $n = 1/2$ for direct allowed transitions.

The optical band gap corresponds to 2.750 eV for the $g\text{-C}_3\text{N}_4$ bulk and 2.625 eV for the C_3N_4 nanosheets. The narrowing of the optical band gap could be due to various reasons such as the presence of oxygen vacancies and the surface interaction between $g\text{-C}_3\text{N}_4$ nanosheet and H_2S or CS_2 .

The emission peak of $g\text{-C}_3\text{N}_4$ bulk is around 465 nm, whereas the position of the emission peak shows a shift to about 444 nm for $g\text{-C}_3\text{N}_4$ nanosheets, consistent with the conditions observed in the UV-vis spectroscopy. The luminescence (PL) spectra of the samples were examined at an excitation wavelength of 340 nm and are presented in Figure 8. The $g\text{-C}_3\text{N}_4$ bulk shows very strong emission peaks centered at 437 nm, turning to 454 nm in the nanosheet. This is consistent with the shift of the absorption band edge after calcination and may be due to the high degree of condensation and filling between the layers in the structure. The PL intensity of the $g\text{-C}_3\text{N}_4$ nanosheet is lower than the $g\text{-C}_3\text{N}_4$ bulk due to the inhibition of the recombination rate of photo-excited electron-hole pairs, which in turn is ascribed to the thin structure of the few-layer $g\text{-C}_3\text{N}_4$. This inhibition in the recombination of charge carriers is helpful for the enhancement of photocatalyst performance [22, 23].

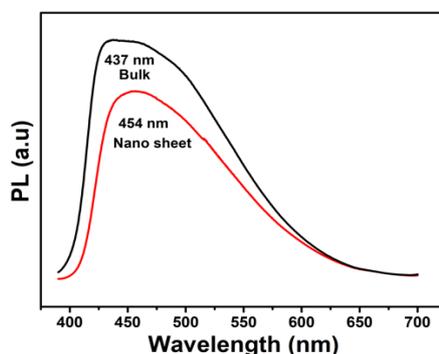


Figure 8. PL spectra of $g\text{-C}_3\text{N}_4$ bulk and nanosheet.

Lorentz fitting of the PL peaks helps us obtain a clear understanding of the nature and origin of excitons in the g-C₃N₄ bulk and nanosheet. The R² value of fit Lorentz results is greater than 0.97, which is very reliable. Three major emission centers have been demonstrated in the fitting and decomposition of the emission spectrum of the g-C₃N₄ samples. Figure 9(a) shows the line shape analysis of the g-C₃N₄ bulk, which includes the emission center P1 (430 nm, 2.89 eV), P2 (460 nm, 2.70 eV), and P3 (519 nm, 2.39 eV). In Figure 9(b), the emission center P1 (439 nm, 2.83 eV), P2 (470 nm, 2.64 eV), and P3 (511 nm, 2.43 eV) for g-C₃N₄ nanosheet is also presented. According to the previous PL study of g-C₃N₄, the optical bandgap states of g-C₃N₄ consist of a sp³ C-N σ band, sp² C-N π band and the lone pair (LP) state of the bridge nitride atom, and the P1, P2 and P3 origin from the 3 different pathways of transitions: π*-π, σ*-LP and π*-LP, respectively [24]. The emission center of the g-C₃N₄ nanosheet has a shorter shift. This shift can be explained by the extension of the g-C₃N₄ network. When more heptazine is connected by the amino group, the π states will hybridize into a broad state, causing the bandgap narrowing of the sp² C-N clusters.

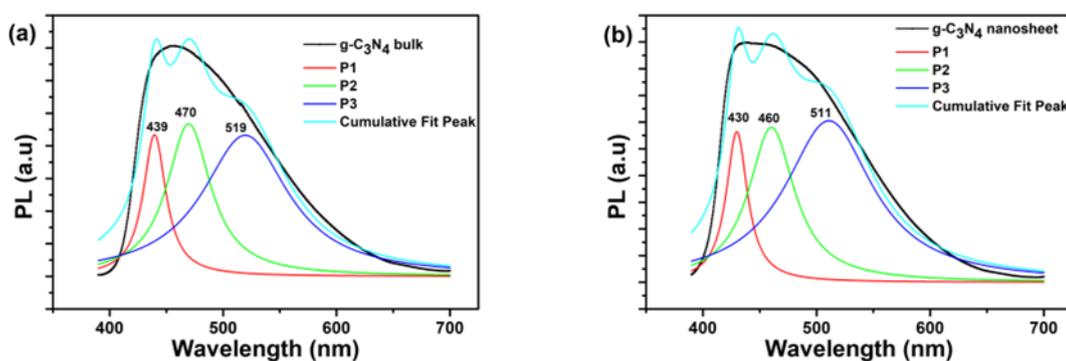


Figure 9. Lorentz fitting of PL emission spectra of the g-C₃N₄ bulk (a) and nanosheet (b) which indicate 3 major PL peaks (P1, P2 and P3).

4. CONCLUSIONS

Polymeric g-C₃N₄ bulk was successfully synthesized with a facile and environmentally benign approach by directly treating low-cost thiourea in air at 550 °C/2 h with a rate of 5 °C/min. Thiourea is a better precursor for the synthesis of g-C₃N₄ bulk than a toxic precursor such as dicyandiamide. The g-C₃N₄ bulk has an optical band gap of around 2.750 eV, while the g-C₃N₄ nanosheet has a narrowing optical bandgap of about 2.625 eV, suitable for visible light utilization. The g-C₃N₄ bulk shows very strong emission peaks centered at 437 nm, turning to 437 nm in the nanosheet. The PL intensity of the g-C₃N₄ nanosheet is lower than the g-C₃N₄ bulk due to the inhibition of the recombination rate of photo-excited electron-hole pairs. This work demonstrates a highly valuable facile method to synthesize high-performance g-C₃N₄ polymeric photocatalysts from easily available thiourea for large-scale environmental and energy applications.

Acknowledgements. We would like to acknowledge the research funding from the Ministry of Education and Training (Grant number: B2023-BKA-04).

Credit authorship contribution statement. Luu Thi Lan Anh: Conceptualization, Supervision, Funding acquisition, Resources. Nguyen Quang Truong: Data curation, Methodology, Investigation, Writing-

Original Draft, Nguyen Thi Tuyet Mai: Methodology, Investigation, Formal analysis. Nguyen Cong Tu: Formal analysis, Data curation, Methodology. Le Manh Cuong: Supervision, Writing - Review & Editing.

Declaration of competing interest. The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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